



The construction of nanotubular Ni-Mg phyllosilicate solid solution catalyst for boosting CO₂ catalytic conversion: Identifying the species variations and interactions

Dehui Wang^a, Jia Liu^a, Hai Li^a, Qing Liu^{a,*}, Yu Cheng^{b,*}, Xing Fan^{a,*}, Peng Liang^{a,*}

^a Key Laboratory of Low Carbon Energy and Chemical Engineering, Shandong University of Science and Technology, Qingdao 266590, China

^b College of Transportation, Shandong University of Science and Technology, Qingdao 266590, China

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ABSTRACT

Addressing the problems of easy collapse, Ni sintering and low catalytic activity over the nanotubular nickel phyllosilicate catalyst, the Mg species was doped via a simple hydrothermal method to construct the nanotubular Ni-Mg bimetallic phyllosilicate solid solution. The high thermal stability, nonreducible property and strong basicity of magnesium species led to the enhancement of anti-collapse, Ni-sintering resistance and basicity. As a result, the fine Ni particles and enhanced H₂ and CO₂ chemisorption and activation properties resulted in boosting the catalytic activity of both CO₂ methanation and CH₄-CO₂ reforming reactions. The Ni/Mg ratios and the reduction temperatures were detailed optimized, and the 600°C-reduced Ni₈Mg₂-NT catalyst with the Ni/Mg ratio of 8:2 was the optimal, which also exhibited high long-term stability. In short, the formed Ni-Mg-phyllosilicate solid solution structure and the special interactions of different species brought about high catalytic performance of the nanotubular Ni-Mg phyllosilicate catalyst.

1. Introduction

Over the last few decades, massive CO₂ emissions deriving from the fossil fuels consumption have led to serious environmental problems including global warming [1], ocean acidification, as well as extreme climate change [2,3], which made the sustainable development of the economy and environment at high risk and even threatened human survival [4]. Catalytic conversion of CO₂ to the valuable chemicals is an important form to utilize CO₂ as a resource [5], among which CO₂ methanation and CH₄-CO₂ reforming have drawn much attention due to the obtaining methane or syngas with high efficiency [6]. The two reactions should be performed in the presence of a catalyst; coincidentally, the nickel-based catalysts are active for the both reactions, because of the requirement of the similar catalytic active centers deriving from metallic Ni [7,8]. Furthermore, the two reactions are usually performed at high temperatures for economic reasons, together with the existence of hydrocarbon, the potential sintering of nickel particles and deposited carbon are the challenges for these catalysts [9].

Ni-phyllosilicate is a layered silicate consisting of continuous Si-O tetrahedral structural units alternating with octahedral structural units consisting of nickel oxide or hydroxide [10], which can be converted to

Ni/SiO₂ after reduction. The strong metal-support interaction originated from nickel phyllosilicate is the obvious advantage of the obtained Ni/SiO₂ catalyst compared with the conventional one prepared by precipitation or impregnation method [11]. Thus, Ni-phyllosilicate has been widely used in various catalytic reactions operated at high temperatures. For instance, Yang et al. synthesized Ni-phyllosilicate on the surface of SiO₂ microspheres by the evaporative ammonia method and obtained high resistance to sintering and coking in the partial oxidative reforming reaction of CH₄ [12]. Kong et al. used silica sol and nickel nitrate as precursors to produce Ni nanoparticle-embedded nickel phyllosilicate catalysts by the evaporative ammonia method, which showed high low-temperature activity (130–150 °C) in the hydrogenolysis reaction of 5-hydroxymethylfurfural [13]. Authors also reported that the unreduced nickel phyllosilicate species could provide the Lewis acid center. Das et al. prepared CeO₂-modified nickel phyllosilicate catalysts with a sandwich-like structure using silica nanospheres as a template, and this domain-limited structure and the auxiliary effect of CeO₂ allowed the catalysts to be stable at 600 °C for 72 h with excellent resistance to carbon deposition for the dry reforming reaction of biogas [14].

Although different types of nickel phyllosilicate have been widely

* Corresponding authors.

E-mail addresses: qliu@sdust.edu.cn (Q. Liu), yicheng1979@163.com (Y. Cheng), fanxing@sdust.edu.cn (X. Fan), liangpeng202@hotmail.com (P. Liang).

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studied, the morphology of nickel phyllosilicate also produce an important effect on the catalytic performance. Different from the conventional layered Ni-phyllosilicate, the nanotubular nickel phyllosilicate obtains the special morphology with the characteristics of low mass transfer resistance, which can also improve the reaction rate of the catalyst to a certain extent. In addition, the unique nanotubule-like structure can increase the specific surface area resulting in the high Ni dispersion after reduction. Guo et al. designed a series of nickel phyllosilicate nanotube catalysts, in which the interfacial reaction between CeO_2 and phyllosilicate produced a large number of extrinsic defects as the active sites for methane dry reforming [15]. Although the tubular nickel phyllosilicate has displayed many advantages and been used in different fields, it also suffered from the problems of easy collapse and Ni sintering leading to the poor catalytic activity especially for the reactions carried out at high temperatures including CO_2 methanation [16] and $\text{CH}_4\text{-CO}_2$ reforming [17]. Therefore, the modification of the tubular nickel phyllosilicate to improve the anti-collapse and anti-sintering properties is in urgent need.

As one of the key components of a heterogeneous catalyst, the promoter can change the interaction between active components and the coordination environment, and enhance the chemisorption and activation of reactant molecules, resulting in the improved catalytic performance [18–20]. MgO is a widely used structure promoter [21], and its strong basicity is a promising advantage for CO_2 chemisorption and activation [22–24]. Jing et al. proposed the Ni-MgO/ SiO_2 catalysts with adjustable Ni particle size for $\text{CH}_4\text{-CO}_2$ reforming with high coking resistance [25]. Additionally, the Ni- $\text{SiO}_2\text{-MgO}$ catalyst with high basicity exhibited the high structural stability and good catalytic activity for reforming reactions [21]. The Ni/MgO catalysts with different Ni/Mg ratio were synthesized by coprecipitation method for sorbitol hydrolysis tests, whose activity was closely related to the Ni surface area and alkalinity deriving from MgO [26]. Zhao et al. reported that synergistic interaction of MgO and NiO nanoparticles over the Ni-MgO/MAS-24 catalyst made the Ni particles highly dispersed and effectively prevented coking and aggregation of the active species [27]. In addition, the multicore-shell Ni-Mg phyllosilicate nanotubes@silica catalyst was synthesized with high activity for $\text{CH}_4\text{-CO}_2$ reforming, where the silica shell helped to enhance the thermal stability and suppress coking formation [28]. In summary, the addition of Mg species is a reasonable strategy to boost the activity and stability of the catalyst especially for the reactions with CO_2 as raw material.

In this work, a group of tubular nickel-magnesium bimetallic phyllosilicate catalysts were synthesized, which were tuned with different Ni/Mg ratios (Ni-NT, $\text{Ni}_8\text{Mg}_2\text{-NT}$, $\text{Ni}_5\text{Mg}_5\text{-NT}$, $\text{Ni}_2\text{Mg}_8\text{-NT}$, Mg-NT) and evaluated for both CO_2 methanation and $\text{CH}_4\text{-CO}_2$ reforming reactions. The reduction temperature was also investigated to further improve the catalytic activity, and the results showed that the $\text{Ni}_8\text{Mg}_2\text{-NT}$ catalyst reduced at 600°C was the most competitive one. In order to reveal the Mg species variation during the synthesis, calcination and reduction process as well as the interaction between Mg species and Ni active centers, various tests and characterizations were performed, which were shown in the following sections.

2. Experimental section

2.1. Materials

All materials including chemicals and gases were displayed in Appendix A. Supplementary data in detail.

2.2. Synthesis of nanotubular Ni phyllosilicate

The nanotubular nickel or/and magnesium phyllosilicates were prepared using a hydrothermal method (Fig. 1), and the experimental parameters and naming rule were shown in Table 1 and Table S1, respectively.

Typically, $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (1.16 g, 4 mmol) was added in the deionized water (60.0 mL) to form a green transparent solution. A sodium silicate solution ($0.5\text{ mol}\cdot\text{L}^{-1}$, 8.0 mL) was added to the above solution under stirring, and the light green precipitate was formed. After the further stirring for 10 min, NaOH (6.00 g, 0.15 mol) was added to the above mixture, which was treated in an autoclave at 200°C for 24 h. Finally, the sample was collected after centrifugation, washing, and air drying at 80°C for 10 h, and referred to as “Ni-NT”, in which “NT” stood for “nanotubular phyllosilicate”.

2.3. Synthesis of nanotubular Ni-Mg bimetallic phyllosilicate

The nanotubular Ni-Mg bimetallic phyllosilicates with different Ni/Mg ratio were synthesized using the same procedures with Ni-NT except for the addition of $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (0.93 g, 3.2 mmol) together with the certain amount of $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (Table 1). The obtained products were referred to as “ $\text{Ni}_x\text{Mg}_y\text{-NT}$ ”, in which x/y was the molar ratio of Ni/Mg with the values of 8/2, 5/5 and 2/8. Here, the Ni content of the as-synthesized Ni-NT, $\text{Ni}_8\text{Mg}_2\text{-NT}$, $\text{Ni}_5\text{Mg}_5\text{-NT}$ and $\text{Ni}_2\text{Mg}_8\text{-NT}$ measured by ICP was 46.26, 34.83, 24.74 and 10.12 wt%, respectively (Table 1).

2.4. Synthesis of nanotubular magnesium phyllosilicate

The procedures of the preparation of nanotubular magnesium phyllosilicate were almost the same with those of Ni-NT except for the addition of $\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (1.02 g, 4 mmol) rather than Ni ($\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$. The obtained product was referred to as “Mg-NT”.

Table 1

Experimental parameters for the preparation of Ni-NT, $\text{Ni}_x\text{Mg}_y\text{-NT}$ (x/y = 8/2, 5/5 and 2/8) and Mg-NT.

Samples	Ni ($\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (mmol)	Mg ($\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (mmol)	Na_2SiO_3 (mL) ^a	NaOH (mol)	Ni content (wt%) ^b
Ni-NT	4.0	—	8.0	0.15	46.26
$\text{Ni}_8\text{Mg}_2\text{-NT}$	3.2	0.8	8.0	0.15	34.83
$\text{Ni}_5\text{Mg}_5\text{-NT}$	2.0	2.0	8.0	0.15	24.74
$\text{Ni}_2\text{Mg}_8\text{-NT}$	0.8	3.2	8.0	0.15	10.12
Mg-NT	—	4.0	8.0	0.15	—

^a $\text{Na}_2\text{SiO}_3 \cdot 9\text{H}_2\text{O}$ aqueous solution of $0.5\text{ mol}\cdot\text{L}^{-1}$.

^b Ni content, determined by ICP.

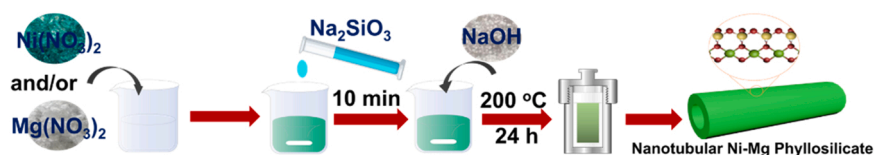


Fig. 1. Schematic diagram of the typical catalyst synthesis using a hydrothermal method.

2.5. Characterizations

The characterizations including inductively coupled plasma atomic emission spectrometry (ICP), X-ray diffraction (XRD) analysis, scanning electron microscope (SEM) and transmission electron microscope (TEM) observations, N_2 sorption, Fourier-transform infrared spectra (FT-IR), H_2 temperature programmed reduction (H_2 -TPR), H_2 temperature programmed desorption (H_2 -TPD), CO_2 temperature programmed desorption (CO_2 -TPD), active surface area (S_m), Ni dispersion (D) measurement, X-ray photoelectron-spectrometer (XPS), and *in situ* diffuse reflectance infrared Fourier transform spectroscopy (*in situ* DRIFTS) were displayed in Appendix A. [Supplementary data](#) in detail.

2.6. Catalytic performance measurements

The catalytic tests procedures including catalytic activities for CO_2 methanation and CH_4 - CO_2 reforming, activation energy (E_a), turnover frequency (TOF) measurements, and long-term stability were shown in Appendix A. [Supplementary data](#) in detail.

3. Results

3.1. Catalytic activity test

Firstly, the four 700°C-reduced Ni-phyllsilicate-based catalysts were evaluated for CO_2 methanation. In Fig. 2a–c, the CO_2 conversion of Ni-NT showed a volcanic trend with the maximum at around 450 °C. As well known, CO_2 methanation reaction was strongly exothermic, and high temperatures were not conducive to this reaction according to Le Chatelier's principle. Interestingly, the three Ni_xMg_{3-x} -NT catalysts

showed the similar change rule with Ni-NT. In addition, the CH_4 selectivity of Ni_xMg_{3-x} -NT was relatively close with each other, which was higher than that of Ni-NT. The only by-product detected by gas chromatography was CO originated from the reverse water gas shift reaction. Among them, Ni_8Mg_2 -NT was the best with the maximum CO_2 conversion of 72.5% and CH_4 yield of 72.1% at 450 °C, followed by Ni_5Mg_5 -NT > Ni-NT > Ni_2Mg_8 -NT (Fig. 2a–c and Fig. S1a). Apparently, compared with Ni/NT, the construction of nanotubular Ni-Mg bimetallic phyllosilicate with the appropriate Ni/Mg ratio could improve the catalytic activity. For Ni_2Mg_8 -NT, the lowest Ni content should be the main reason for the worst catalytic activity [29]. However, the CO_2 conversion could not reach the thermodynamics equilibrium curve in the whole temperature range, indicating the catalytic activity of the 700 °C-reduced catalyst was relatively low.

In order to further boost the catalytic activity, the reduction temperature of 600 °C was also investigated. Here, only the three catalysts of Ni-NT, Ni_8Mg_2 -NT and Ni_5Mg_5 -NT with good performance were selected for evaluation. For the three 600 °C-reduced catalysts, the catalytic activity followed similar volcanic trends to those of the 700 °C-reduced catalysts (Fig. 2d–e). Significantly, the higher catalytic performance was obtained; in detail, the CO_2 conversion of Ni_8Mg_2 -NT could reach thermodynamics equilibrium at 450 °C, and its CH_4 selectivity almost reached the thermodynamics equilibrium from the initial temperature of 300 °C and exceeded the equilibrium curve due to the control of both thermodynamics and dynamics at high temperatures. As a result, at 450 °C, Ni_8Mg_2 -NT, which was reduced at 600 °C, showed a maximum CO_2 conversion of 78.8% and CH_4 yield of 77.2%. In addition, the catalytic performance of other catalysts also increased, indicating that the low temperature reduction led to higher catalytic activity.

On the contrary, the activities of the three 500 °C-reduced catalysts

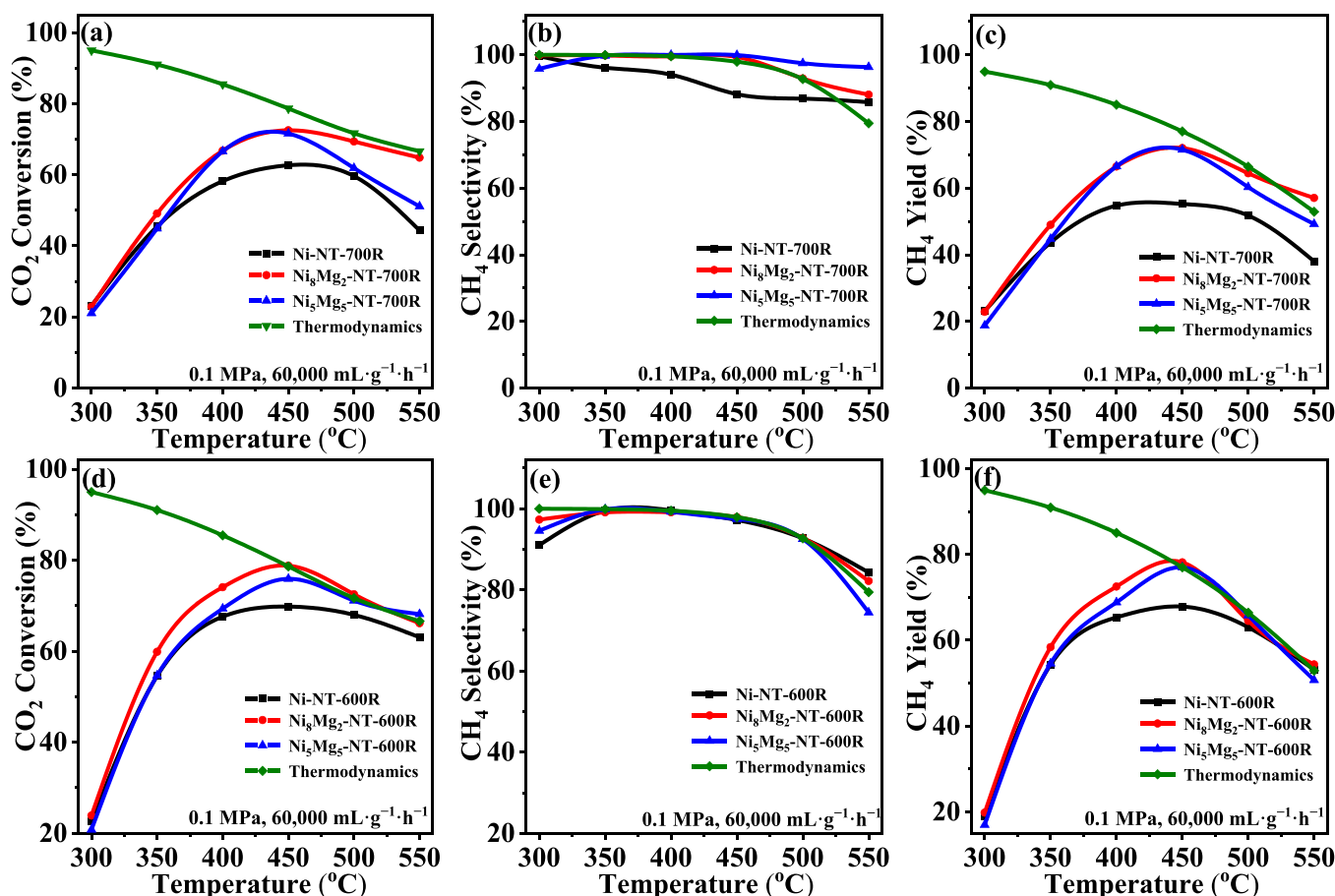


Fig. 2. Catalyst activity of the 700 or 600°C-reduced catalysts for CO_2 methanation: (a and d) CO_2 conversion, (b and e) CH_4 selectivity, and (c and f) CH_4 yield.

were all very low because of the low reduction degree (Fig. S1b). Surprisingly, the CH₄ selectivity was also decreased compared with that of 700 or 600 °C-reduced catalyst (Fig. S1c); as a result, the CH₄ yield was also very low (Fig. S1d). Despite all this, Ni₈Mg₂-NT was the best among the three 500 °C-reduced catalysts. Furthermore, these catalysts were also reduced at 400 °C for 1 h; unfortunately, the 400 °C-reduced catalysts were inactive for CO₂ methanation because of the too low reduction temperature (data not shown here). At the same time, the CO selectivity of all samples was shown in Fig. S1e. Apparently, the above results indicated that reduction temperature was an important parameter for investigation of catalytic activity, and 600 °C was the optimal value in this work.

Furthermore, the three catalysts (Ni-NT, Ni₈Mg₂-NT and Ni₅Mg₅-NT) were also reduced at the optimal temperature of 600 °C and evaluated for CH₄-CO₂ reforming (Fig. 3). This was an endothermic reaction and the catalytic performance of the three catalysts including the conversion of CO₂ and CH₄, H₂ selectivity as well as H₂ yield increased in the operation temperature range of 550 – 800 °C. Overall, Ni₈Mg₂-NT was still the best with the maximum CO₂ conversion of 95.8%, CH₄ conversion of 97.9%, and H₂ yield of 97.9% at 800 °C, followed by Ni₅Mg₅-NT, and Ni-NT was the worst. To sum up, combined with the above evaluation results, Ni₈Mg₂-NT exhibited the best catalytic performance for both CO₂ methanation and CH₄-CO₂ reforming reactions in this work.

3.2. X-ray diffraction analysis

The as-synthesized Ni-NT catalyst showed diffraction peaks corresponding to Ni₃Si₂O₅(OH)₄ (JCPDS No. 20–0791) (Fig. 4a) [30], suggesting that sodium silicate and nickel nitrate had completely converted to nickel phyllosilicate after the hydrothermal treatment at 200 °C for 24 h. After the doping of Mg species, the XRD diffraction patterns of Ni₈Mg₂-NT, Ni₅Mg₅-NT and Ni₂Mg₈-NT (shown in Fig. S2b) were almost the same with that of Ni-NT. There might be two existence states of Mg species: the one was in the form of magnesium phyllosilicate to construct the mixture of Ni-phyllosilicate and Mg-phyllosilicate; the other was in the form of Ni-Mg phyllosilicates solid solution with high Mg dispersion [31]. In order to investigate the two possibilities, the as-synthesized Mg-NT sample was also analyzed and it showed the similar diffraction peaks corresponding to Mg₃Si₂O₅(OH)₄ (JCPDS No. 27–1275). Therefore, it was reasonable that the five samples above with the different components obtained the similar crystalline phase structure with the ratio of SiO₄ tetrahedron to M-O(OH) octahedron (M = Ni and/or Mg coordinated to six O atoms or OH groups) of 1:1. After the careful matching with the PDF card, it was found that Ni_xMg_y-NT belonged to Ni-Mg phyllosilicates solid solution with the formula of [Ni_{1-x}Mg_x]₃(OH)₄Si₂O₅·zH₂O, which was also a metal phyllosilicate with 1:1 type. In short, Ni-Mg phyllosilicates should be in the state of solid solution rather than physical mixture of Ni-phyllosilicate and Mg-phyllosilicate (Fig. S2c).

As mentioned in Figs. 2 and 3 as well as Fig. S1, the different reduction temperature could affect the catalytic performance; thus, the

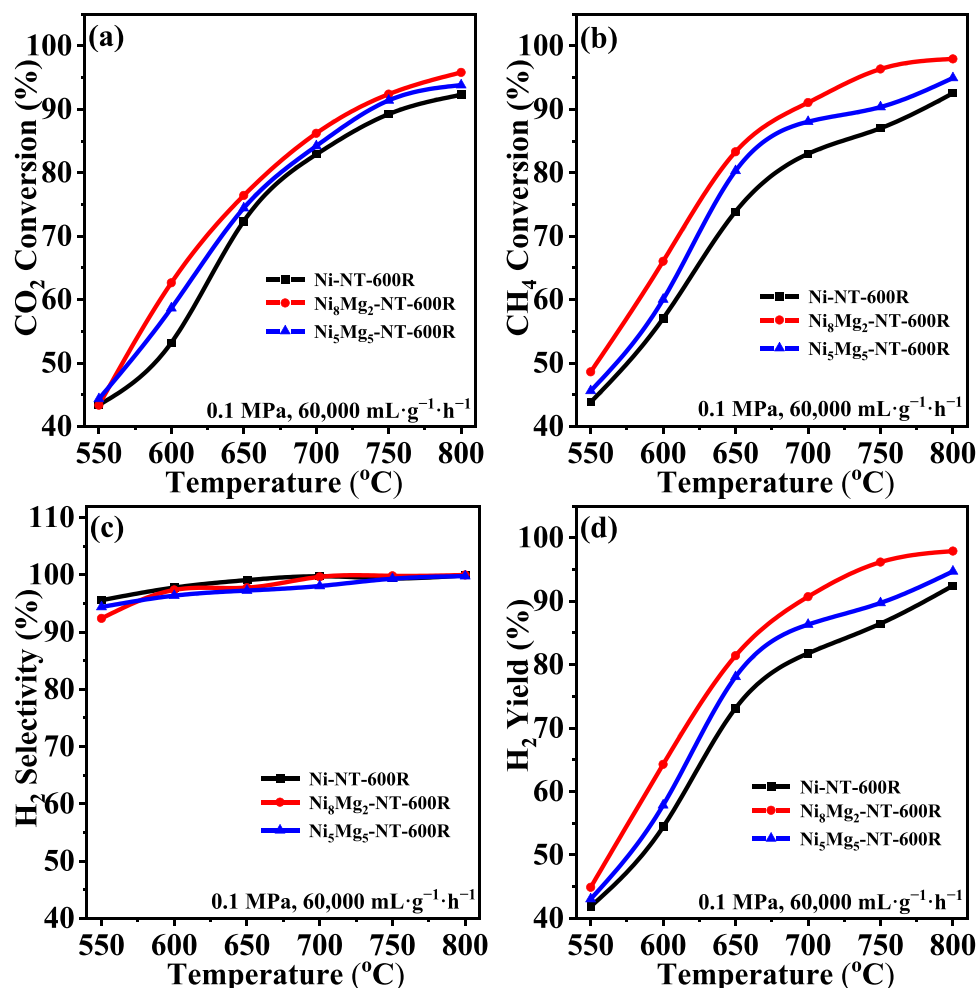


Fig. 3. Catalyst activity of the 600°C-reduced catalysts for CH₄-CO₂ reforming: (a) CO₂ conversion, (b) CH₄ conversion, (c) H₂ selectivity, and (d) H₂ yield.

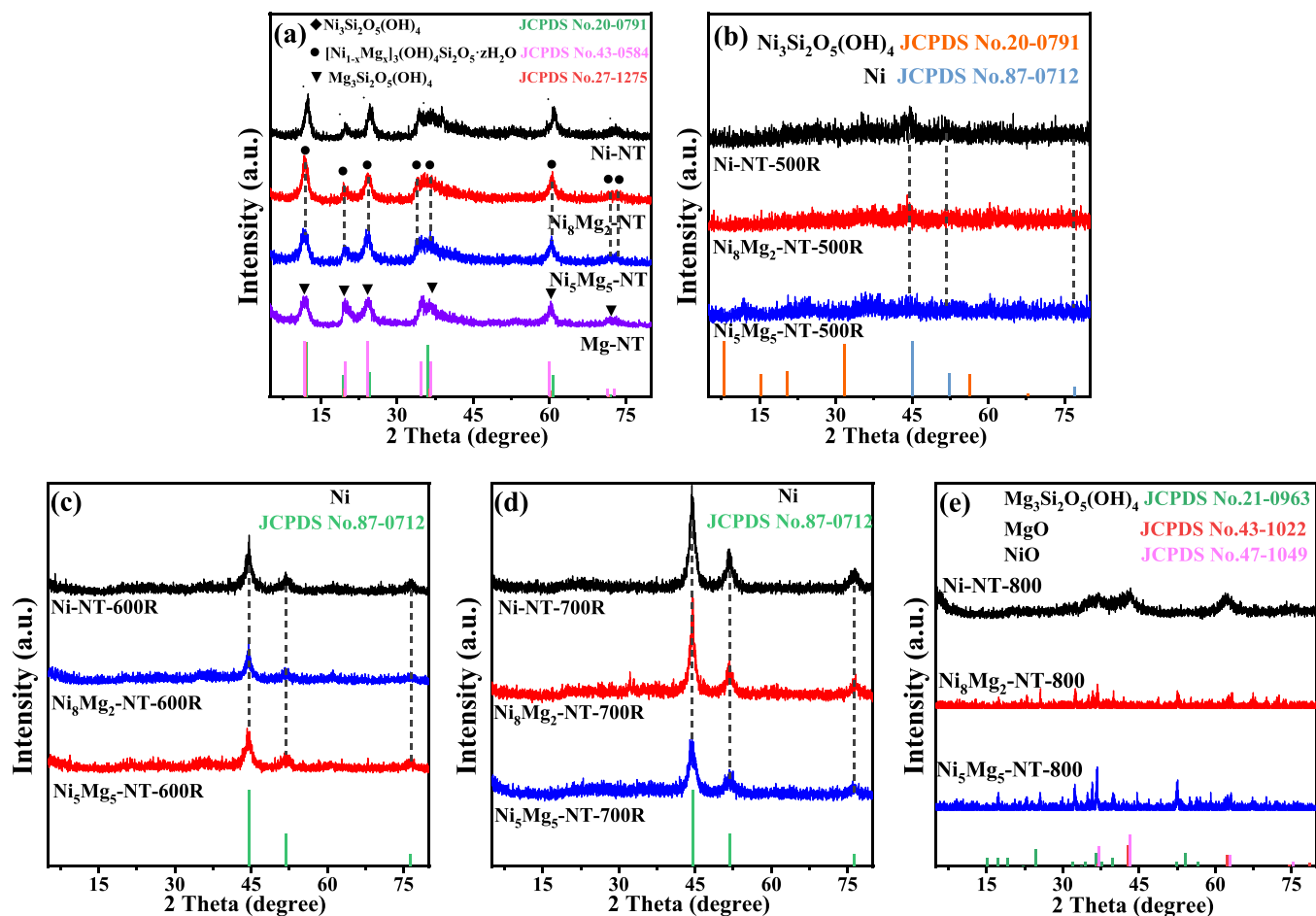


Fig. 4. XRD patterns: (a) the as-synthesized catalysts, (b) the 500°C-reduced catalysts, (c) the 600°C-reduced catalysts, (d) the 700°C-reduced catalysts, (e) the 800°C-calcined catalysts.

reduced Ni-NT, $\text{Ni}_8\text{Mg}_2\text{-NT}$, and $\text{Ni}_5\text{Mg}_5\text{-NT}$ catalysts at 500, 600 and 700 °C were also analyzed by XRD. Surprisingly, after the reduction at 500 °C, the diffraction peaks attributing to Ni-phyllsilicate disappeared, and a tiny peak occurred at 44.4° on the XRD pattern of 500°C-reduced Ni-NT belonging to metallic Ni (JCPDS No. 87–0712) (Fig. 4b). On the contrary, there was no obvious diffraction peak attributing to Ni-phyllsilicate or Mg-phyllsilicate over the 500°C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ and $\text{Ni}_5\text{Mg}_5\text{-NT}$, and the peaks of metallic Ni could also not be identified. After the reduction at 600 °C, the three diffraction peaks were obviously observed on the all samples, which were attributed to the metallic nickel (Fig. 4c) [26]. Similar with Fig. 4b, there was no peak attributing to metal phyllsilicate or Mg species. Furthermore, for the 700 °C-reduced samples, the similar three diffraction peaks with stronger intensity were observed attributing to Ni (Fig. 4d) [26]. It was noteworthy that these diffraction peaks became sharper and narrower as the reduction temperature increased, suggesting the increase of the Ni crystal size. It was a great pity that the Ni crystal size could not be calculated in Fig. 4b due to the too weak peaks. As expected, the nickel crystal size of the three 600°C-reduced catalysts was 4.84, 4.17, and 4.01 nm, respectively; while that of the three 700°C-reduced catalysts was 5.01, 4.73, and 4.35 nm (Table 2 and Table S2). It was apparent that the overreduction of a catalyst at high temperature led to further growth of Ni particle size resulting in decrease of catalyst activity.

In order to further investigate the change of crystal phase with the increase of reduction temperature, $\text{Ni}_8\text{Mg}_2\text{-NT}$ was picked out as a representative, and the XRD patterns of the as-synthesized as well as the 400, 500, 600 and 700°C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ were shown in Fig. S2a. The XRD pattern of 400°C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ was almost the same with

Table 2
Physical and chemical properties of samples.

Samples	Ni crystal size (nm) ^a	Ni particle size (nm) ^b	S_{BET} ($\text{m}^2\cdot\text{g}^{-1}$) ^c	V_{p} ($\text{cm}^3\cdot\text{g}^{-1}$) ^d	D (nm) ^e
$\text{Ni}_8\text{Mg}_2\text{-NT-400R}$	—	—	136	0.49	11.56
$\text{Ni}_8\text{Mg}_2\text{-NT-500R}$	—	2.61	—	—	—
Ni-NT-600R	4.84	5.12	87	0.41	12.36
$\text{Ni}_8\text{Mg}_2\text{-NT-600R}$	4.17	4.78	112	0.44	12.61
$\text{Ni}_5\text{Mg}_5\text{-NT-600R}$	4.01	4.23	108	0.48	18.03

^a Calculated by the XRD diffraction peak ($2\theta = 44.4^\circ$) using the Debye–Scherrer equation.

^b Estimated from the TEM images.

^c S_{BET} , surface area, derived from the BET equation.

^d V_{p} , pore volume, obtained from the volume of nitrogen adsorbed at the relative pressure of 0.97.

^e D , average pore diameter, derived from BET equation.

the as-synthesized one, indicating that 400 °C was too low to reduce Ni-NT to metallic Ni. A nearly unrecognizable peak occurred at around 44.4° on the XRD pattern of 500°C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ attributing to the (111) crystal plane of the metallic Ni, indicating that the metallic nickel with very small size was reduced from $\text{Ni}_8\text{Mg}_2\text{-NT}$ at 500 °C. As the reduction temperature increased, the diffraction peaks belonging to the metallic Ni became higher and narrower, signifying the increase of the Ni crystal size.

It was very strange that, in Fig. 4b–d and Fig. S2a, no diffraction peaks attributing to Mg phyllsilicate or MgO were found on the XRD patterns of the reduced samples even at the high Mg species content. To

further reveal the variation of Mg species, the as-synthesized Ni-NT, $\text{Ni}_8\text{Mg}_2\text{-NT}$, $\text{Ni}_5\text{Mg}_5\text{-NT}$, and Mg-NT were calcined at 800 °C in air. Interestingly, on the XRD pattern of the Ni-NT catalyst after the calcination at 800 °C, the diffraction peaks of Ni-phyllsilicate disappear and new peaks attributing to NiO with low intensity could be seen (Fig. 4e), indicating that the tubular Ni phyllsilicate was decomposed to NiO and the amorphous silica. For the XRD patterns of $\text{Ni}_8\text{Mg}_2\text{-NT}$ and $\text{Ni}_5\text{Mg}_5\text{-NT}$ after the calcination at 800 °C, the peak intensity also decreased compared with the as-synthesized ones in Fig. 4a. MgO (JCPDS No. 43–1022) as well as NiO (JCPDS No. 47–1049) together with magnesium phyllsilicate (JCPDS No. 21–0963) were observed over the 800 °C-calcined $\text{Ni}_8\text{Mg}_2\text{-NT}$ and $\text{Ni}_5\text{Mg}_5\text{-NT}$ [32]. In addition, the intensities of Mg species increased with the increase of their contents. The above results showed that the nickel phyllsilicate completely decomposed to NiO and SiO_2 , while magnesium phyllsilicate only partially decomposed at 800 °C in air, indicating its higher thermal stability compared with Ni-phyllsilicate. Moreover, the Mg species evolution varied with the operation atmosphere (H_2 or air), which could change its speciation as well as crystal state.

3.3. Morphology analysis

To visualize the morphology of the samples, the SEM images of the as-synthesized samples were shown in Fig. 5. Ni-NT showed a typical nanotube morphology, and the length of the tubes varied in a wide range with the outer diameter of around 25 nm (Fig. 5a), which was the excellent evidence of the successful preparation of nanotubular nickel phyllsilicate. At the same time, the as-synthesized $\text{Ni}_8\text{Mg}_2\text{-NT}$ (Fig. 5b), $\text{Ni}_5\text{Mg}_5\text{-NT}$ (Fig. 5c), $\text{Ni}_2\text{Mg}_8\text{-NT}$ (Fig. S3) and Mg-NT (Fig. 5d) also showed the typical nanotube morphology, which indicated the doping of Mg species did not change the tubular structure. In addition, the length of $\text{Ni}_x\text{Mg}_y\text{-NT}$ and Mg-NT was also unregular, and some of them could be as long as 437 nm; meanwhile, their diameters were almost the same without varying with the composition.

In order to obtain the inner diameter and wall thickness information, the as-synthesized $\text{Ni}_8\text{Mg}_2\text{-NT}$ catalyst was then observed by TEM

(Fig. 6a–c). As expected, the typical tubular structure could be seen in Fig. 6a, whose inner and outer diameter was in the range of 11–12 and 23–25 nm, respectively (Fig. 6b), and the length of the tube could be 321 nm. In addition, the typical cross-sections of nanotubes were spiral or concentric (Fig. 6c). As reported in the literature, the formed $\text{Ni}_3\text{Si}_2\text{O}_5(\text{OH})_4$ nanosheets could be bent along the crystallographic [100] direction and the ends of the rolling sheets met to form a circumference of complete nanotubes during the hydrothermal process (Fig. 6c) [33].

Then, the TEM images of the 400, 500, 600 and 700 °C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ were recorded in Fig. 6d–g. After the reduction at 400 °C, $\text{Ni}_8\text{Mg}_2\text{-NT-400R}$ (Fig. 6d) displayed the similar structure and morphology with the as-synthesized one (Fig. 6a–c). On the contrary, fine Ni particles highly dispersed on the nanotubes were observed over the 500 °C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ (Fig. 6e). As the reduction temperature increased, more and larger black dots could be seen on the TEM image of $\text{Ni}_8\text{Mg}_2\text{-NT}$ after reduction at 600 °C, but the complete tubular structure could still be seen (Fig. 6f). When the reduction temperature reached 700 °C, the tubular morphology was hard to be identified with black dots fully covered the whole catalyst (Fig. 6g). The size of nickel particle size was estimated and that of the $\text{Ni}_8\text{Mg}_2\text{-NT}$ reduced at 500, 600 and 700 °C was 2.61, 4.78 and 5.09 nm respectively (Table 2 and Table S4). Apparently, the high reduction temperature led to the large nickel particle size. In addition, the reduced Ni-NT and $\text{Ni}_5\text{Mg}_5\text{-NT}$ at 600 and 700 °C were also recorded by TEM (Figs. 6g and 6h as well as Fig. S4), which exhibited the similar morphology with the reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$ in Figs. 6f and 6g. The average nickel particle sizes of these catalysts were estimated; in detail, that of the 600 °C-reduced Ni-NT and $\text{Ni}_5\text{Mg}_5\text{-NT}$ catalysts was 5.12 and 4.23 nm, respectively; and those reduced at 700 °C were 5.38 and 4.65 nm (Table 2 and Table S2). Apparently, the doping of Mg species could be conducive to reduce the Ni particle size, and higher Mg content led to smaller Ni particle size owing to its physical barrier effect during reduction process.

Moreover, the elemental distribution was recorded by the dark-field TEM and elemental mapping. For the 600 °C-reduced $\text{Ni}_8\text{Mg}_2\text{-NT}$, the bright spots corresponded the Ni particles in Figs. 7a and 7b; and Ni, Mg,

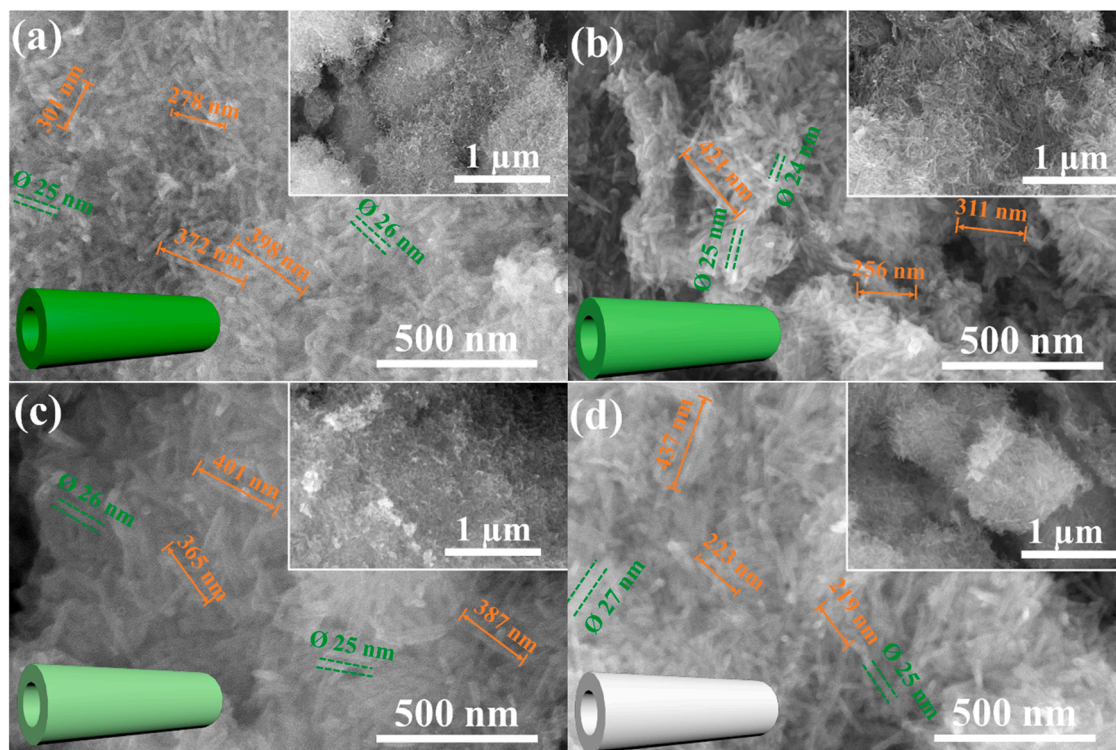


Fig. 5. SEM images of the as-synthesized samples: (a) Ni-NT, (b) $\text{Ni}_8\text{Mg}_2\text{-NT}$, (c) $\text{Ni}_5\text{Mg}_5\text{-NT}$, (d) Mg-NT.

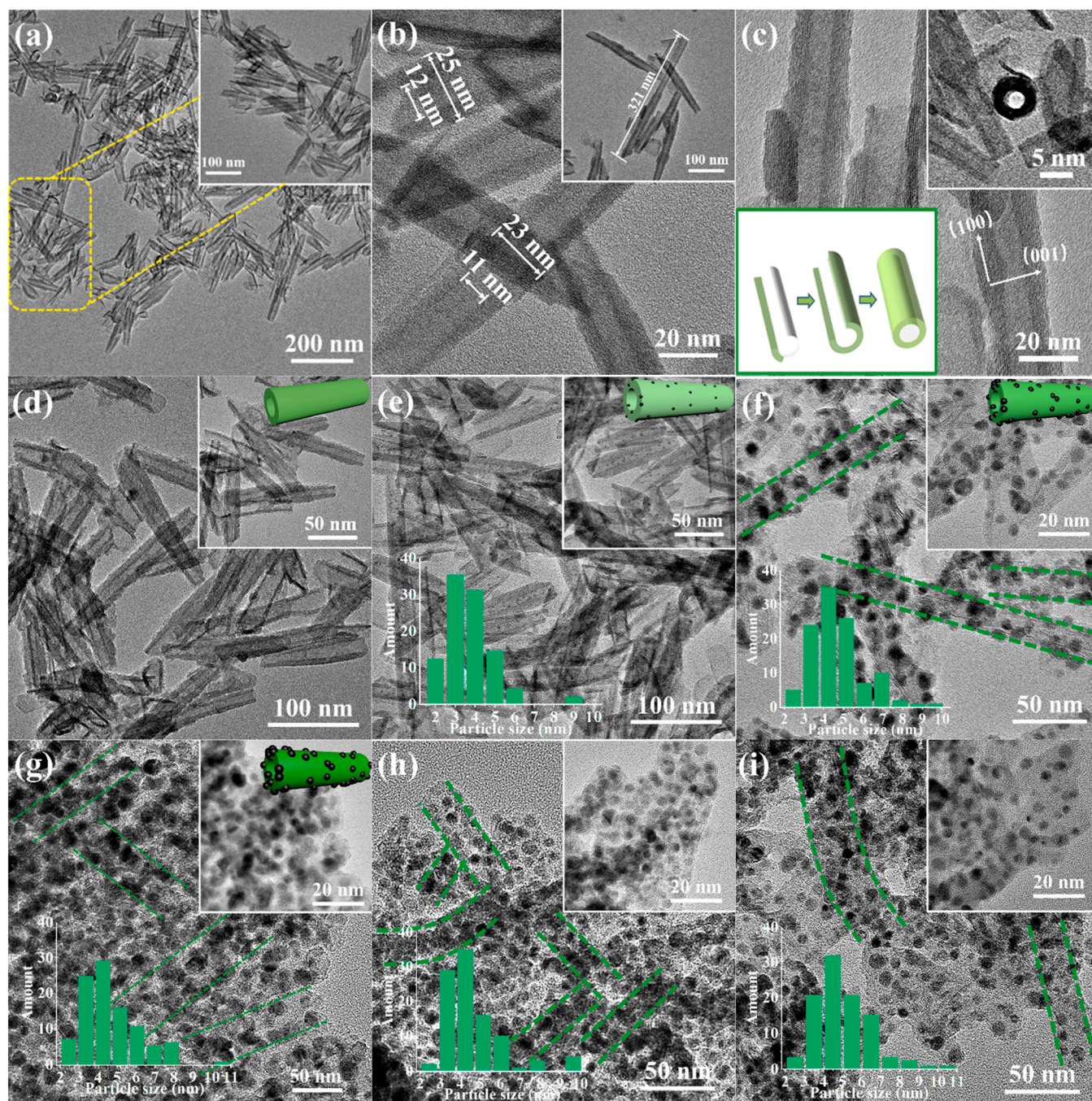


Fig. 6. TEM images of $\text{Ni}_8\text{Mg}_2\text{-NT}$: (a–c) the as-synthesized, (d) 400°C-reduced, (e) 500°C-reduced, (f) 600°C-reduced, and (g) 700°C-reduced; TEM images of the 700°C-reduced Ni-NT (h), and $\text{Ni}_5\text{Mg}_5\text{-NT}$ (i).

O, and Si were uniformly dispersed throughout the catalyst (Fig. 7c–f). As expected, the peaks of Ni and Mg were clearly visible through the EDS analysis (Fig. 7g). In addition, it was observed from the HRTEM images in Fig. 7h that the lattice spacing of around 0.20 nm belonging to the Ni (111) crystal plane could be clearly seen.

Combined with the above results, the variations of Ni-NT and $\text{Ni}_8\text{Mg}_2\text{-NT}$ during the reduction and 800°C-calcination processes were displayed in Fig. 7i. For Ni-NT, the Ni-phyllsilicate ($\text{Ni}_3\text{Si}_2\text{O}_5(\text{OH})_4$) was formed during the synthesis process and converted to Ni/SiO₂ after the reduction above 500 °C. It was noted that the tubular structure was also destroyed during the high-temperature reduction at 700 °C. Meanwhile, the thermal stability of Ni-NT was poor, which completely decomposed to NiO and the amorphous silica after the calcination at

800 °C. On the contrary, for Ni-Mg-NT, the Ni-Mg phyllosilicates ($[\text{Ni}_{1-x}\text{Mg}_x](\text{OH})_4\text{Si}_2\text{O}_5 \cdot z\text{H}_2\text{O}$) solid solution was formed rather than the physical mixture of $\text{Ni}_3\text{Si}_2\text{O}_5(\text{OH})_4$ and $\text{Mg}_3\text{Si}_2\text{O}_5(\text{OH})_4$. During the reduction process, the Ni species showed the similar change with that in Ni-NT, and the metallic Ni could be observed above 500 °C; however, no diffraction peaks attributing to Mg phyllosilicate or MgO were found on the XRD patterns of the reduced samples (500, 600 and 700 °C), indicating the crystal state might convert to the amorphous in the H₂ flow. Surprisingly, MgO and NiO together with magnesium phyllosilicate (JCPDS No. 21–0963) were observed over the 800°C-calcined $\text{Ni}_8\text{Mg}_2\text{-NT}$. Apparently, nickel-magnesium phyllosilicate only partially decomposed during the calcination process at 800 °C in air. In short, the variation of Ni species in Ni-NT and $\text{Ni}_8\text{Mg}_2\text{-NT}$ was the same; while that

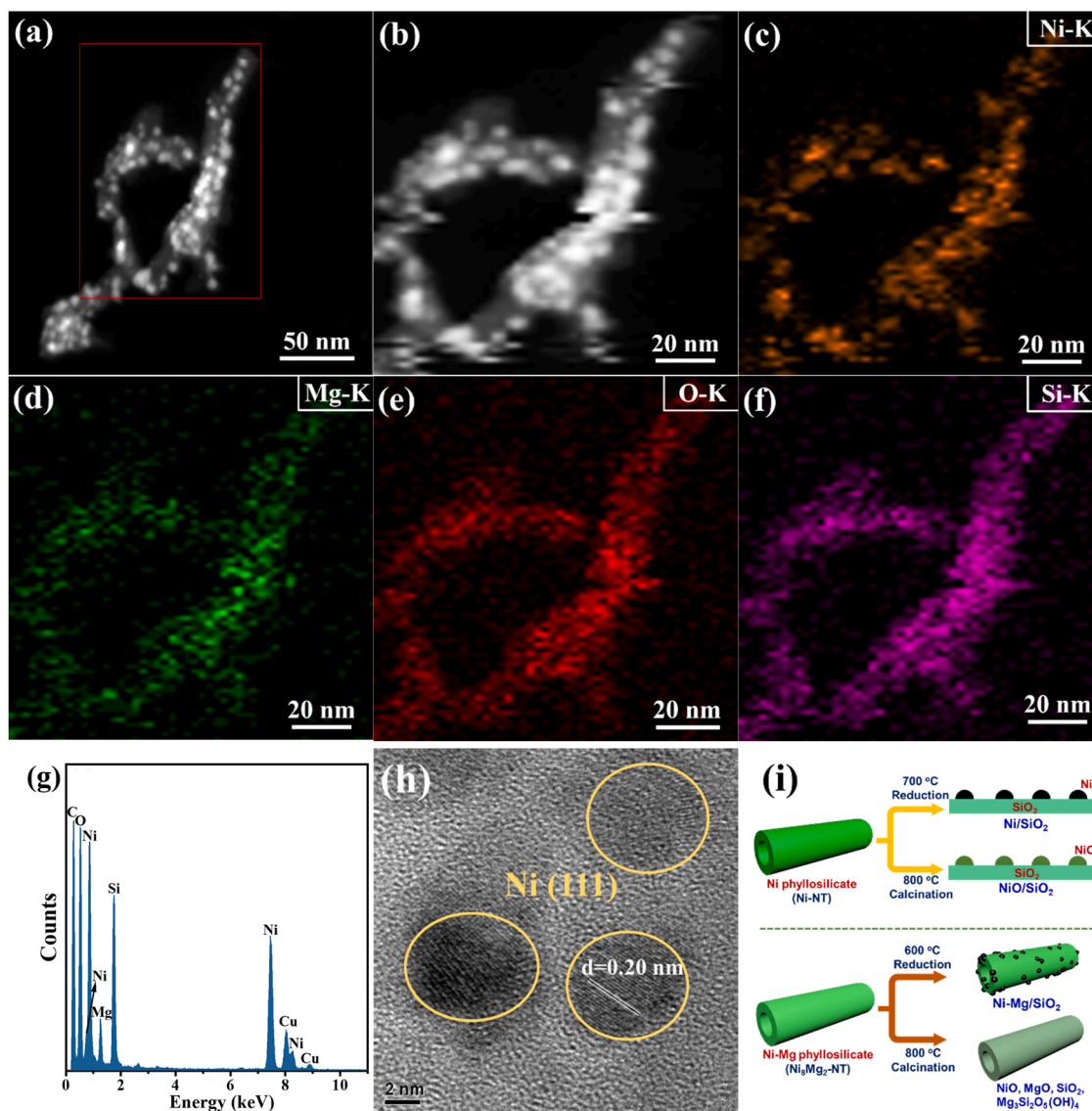


Fig. 7. HAADF-STEM images (a and b) of the 600 °C-reduced Ni₈Mg₂-NT, and elemental maps of Ni (c), Mg (d), O (e), Si (f); EDS analysis (g), HRTEM image of the 600 °C-reduced Ni₈Mg₂-NT (h), and diagram of Ni-NT and Ni₈Mg₂-NT variations during the reduction and 800 °C-calcination processes (i).

of Mg species was different, which varied with the thermal treatment atmosphere (H₂ or air).

3.4. N₂ sorption and FT-IR analyses

In Fig. 8a, the as-synthesized Ni-NT catalyst showed isotherms of type IV with hysteresis loop of H3 between P/P₀ of 0.5 – 0.97, indicating that the tubular nickel phyllosilicate was a typically mesoporous material. Compared to Ni-NT, the adsorption-desorption isotherms of Ni_xMg_y-NT were almost unchanged, suggesting that the addition different amounts of Mg species had little influence on the pore structure (Fig. 8a and S5a). For Mg-NT, the adsorption-desorption isotherms were still isotherms of type IV with hysteresis loop of H3 between P/P₀ of 0.5 – 0.97, which was also a typical mesoporous material. Furthermore, for the as-synthesized Ni-NT, the peak of the pore size distribution (PSD) curve was concentrated around 15 nm (Fig. 8b), which coincided with the TEM observation of the inner diameter. The PSD curves also showed some changes with increase of Mg species; in detail, higher Mg content led to larger pore size. The specific surface areas (*S*_{BET}) and pore volumes (*V*_p) as well as the average pore diameter (*D*_p) of the different as-synthesized catalysts also increased with the increase of Mg content in

Table 3. However, for the three samples (Ni-NT, Ni₅Mg₅-NT and Ni₅Mg₅-NT) reduced at 600 °C, the *S*_{BET} were significantly reduced owing to the partial destruction of pores during the reduction process (Table 2); as a result, there was also some changes for their adsorption-desorption isotherms (Figs. 8c and 8d). Significantly, the obvious changes were observed over the isotherms and the PSD curve of the 600 °C-reduced Ni-NT compared with the as-synthesized one, due to the relatively poor stability and high content of Ni-phyllosilicate.

Next, the functional groups of the as-synthesized Ni-NT and Ni₈Mg₂-NT were also analyzed by FT-IR spectra to reveal the effect of Mg species doping (Fig. S5b). For Ni-NT, the peak at 1093 cm⁻¹ corresponded to the bending and stretching vibrations of the Si–O–Si bond, while the peaks at 3460 and 1630 cm⁻¹ belonged to the stretching and bending vibrations of O–H. Significantly, δ-OH band was also observed at around 670 cm⁻¹, which was the characteristic peak of metal phyllosilicate of 1:1 type [16]. After doping of Mg species, the FT-IR spectrum of Ni₈Mg₂-NT was almost consistent with that of Ni-NT, indicating the modification did not obviously change the functional groups.

In order to detailed investigate the change of pore structure with different reduction temperatures, Ni₈Mg₂-NT was selected as the representative, which was reduced at 400, 500 and 600 °C, respectively

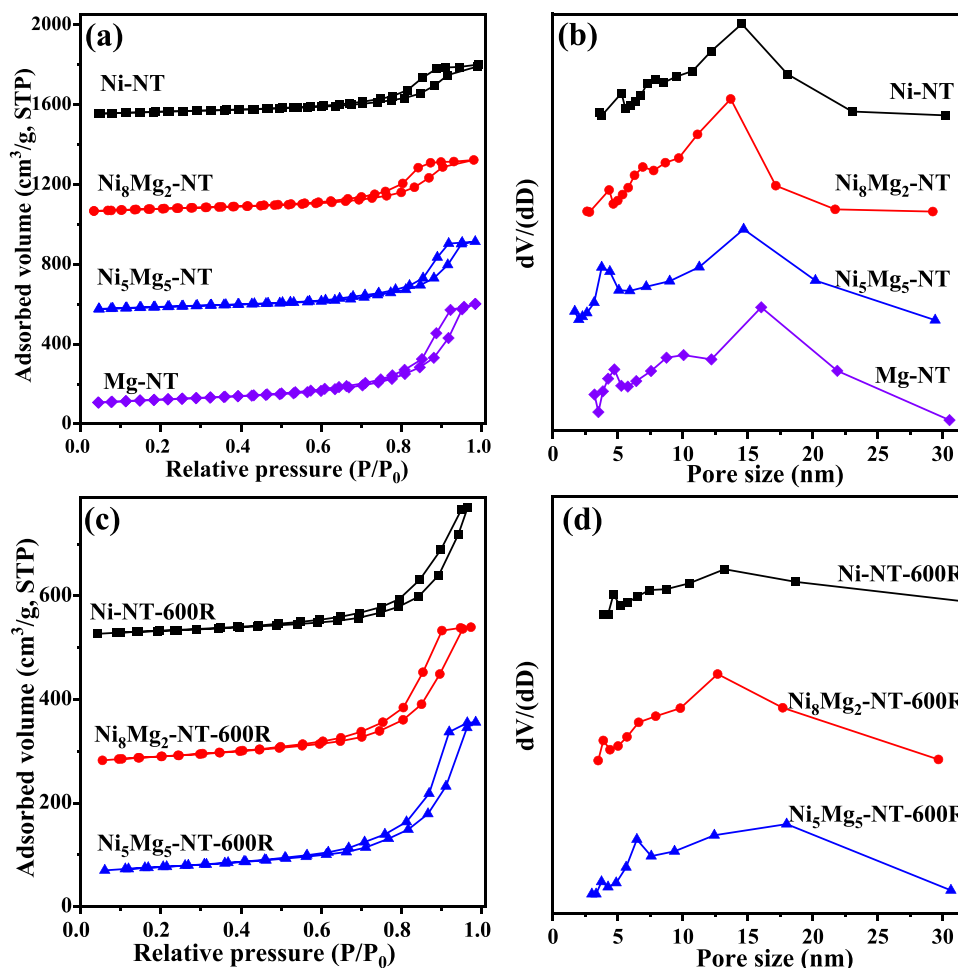


Fig. 8. N_2 adsorption isotherm (a) and the pore size distribution curve (b) of the as-synthesized catalysts; N_2 adsorption isotherm (c) and the pore size distribution curve (d) of 600°C-reduced catalysts.

Table 3

Pore structure parameters of the as-synthesized samples.

Samples	S_{BET} ($m^2 \cdot g^{-1}$) ^a	V_F ($cm^3 \cdot g^{-1}$) ^b	D_p (nm) ^c
Ni-NT	123	0.40	10.72
Ni_8Mg_2 -NT	137	0.41	10.19
Ni_5Mg_5 -NT	145	0.54	13.48
Mg-NT	203	0.80	13.45

^a S_{BET} , surface area, derived from the BET equation.

^b V_F , pore volume, obtained from the volume of nitrogen adsorbed at the relative pressure of 0.97.

^c D_p , average pore diameter, derived from BET equation.

(Fig. S5c and S5d). There was only slight change of the adsorption-desorption isotherms (Fig. S5c) and the PSD curve almost remained unchanged with different treatment (Fig. S5d), indicating the stability of the tubular structure of Ni_8Mg_2 -NT. Moreover, the decrease of S_{BET} and slight increase of D_p pore were also observed with the increase of reduction temperature (Tables 2 and 3).

3.5. Chemisorption analysis

On the H_2 -TPR profile of Ni-NT, the single H_2 consumption peak was in the range of 400 – 900 °C, which centered at around 700 °C (Fig. 9a). For Ni_8Mg_2 -NT and Ni_5Mg_5 -NT, two H_2 consumption peaks at 670 and 774 or 802 °C were observed, attributing to the reduction of nickel species in the Ni-Mg phyllosilicate [12]. Meanwhile, for the peak at

670 °C, the shift towards the low-temperature region after the doping of magnesium species was because the reduction difficulty of the surface Ni phyllosilicate was alleviated [12,21]. On the contrary, the H_2 consumption peak at 700 – 800 °C was attributed to Ni phyllosilicate covered by the Mg phyllosilicate, which increased the reduction difficulty of nickel species [21]. Significantly, the integral areas of the peaks in Fig. 9a were directly proportional to their Ni contents in Table 1. For Mg-NT, the absence of the H_2 consumption peak indicated that Mg^{2+} species could not be reduced to metallic Mg.

Next, the H_2 -TPR analysis of the Ni_8Mg_2 -NT catalyst after reduction at 400, 600, and 700 °C was carried out. For the catalyst reduced at 400 °C, only a single H_2 consumption peak appeared at around 680 °C. Apparently, the high temperature contributed to the reduction of Ni-phyllosilicate [34], which was consistent with the XRD result in Fig. 4a. In contrast, on the H_2 -TPR profiles of the catalysts reduced at 600 and 700 °C, two H_2 consumption peaks appeared in the range of 100 – 300 °C. The one located in the low-temperature region (158 or 168 °C) was attributed to the fine NiO particles; the one centered at 254 or 274 °C should correspond to large NiO particles inside the bulk [35]. In addition, the Ni reduction degrees of the three samples were also calculated, which was in the order of 10.1% (Ni_8Mg_2 -NT-400R) < 96.6% (Ni_8Mg_2 -NT-600R) < 98.2% (Ni_8Mg_2 -NT-700R). Combined with the results in Fig. 2 and Fig. S1b–d, it was seen that high Ni reduction degree could not lead to high catalytic activity, and the reasons would be analyzed in the Discussion section in detail.

Fig. 9c showed the H_2 -TPD profile of the Ni-NT, Ni_8Mg_2 -NT and Ni_5Mg_5 -NT catalysts reduced at 600 °C. Overall, there was only one H_2

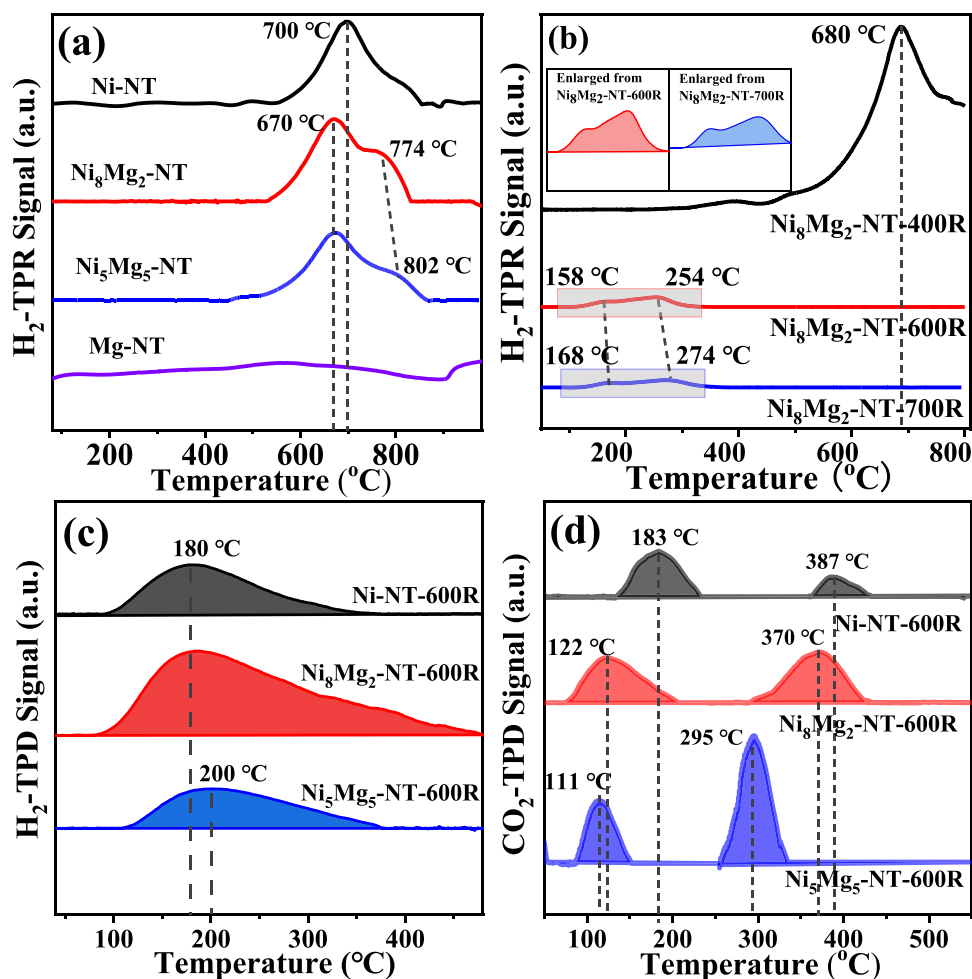


Fig. 9. H₂-TPR profiles of as-synthesized catalysts (a) and Ni₈Mg₂-NT reduced at different temperatures (b), H₂-TPD (c) and CO₂-TPD profiles (d) of the 600 °C-reduced catalysts.

desorption peak centered in the range of 180 – 200 °C. This was attributed to the adsorption of hydrogen from the highly dispersed Ni nanoparticles [36]. Interestingly, although with the lower Ni content, the peak area of Ni₈Mg₂-NT was much larger than that of Ni-NT owing to its much smaller Ni particle size. In addition, the small peak area of Ni₅Mg₅-NT derived from its low Ni content. In all, the H₂ uptakes of the catalysts followed the sequence of Ni₈Mg₂-NT (425.4 μmol·g⁻¹) > Ni₅Mg₅-NT (357.6 μmol·g⁻¹) > Ni₅Mg₅-NT (255.6 μmol·g⁻¹). Furthermore, the calculated Ni dispersion of Ni-NT, Ni₈Mg₂-NT, and Ni₅Mg₅-NT was 8.1%, 16.5%, and 10.3%, respectively (Table 4), indicating that the addition of an appropriate amount of Mg species was conducive to improving the dispersion of Ni. Meanwhile, the active surface areas (*S_m*) of the 600 °C-reduced Ni-NT, Ni₈Mg₂-NT, and

Ni₅Mg₅-NT were also calculated, which was 5.2, 19.1, and 12.5 m²·g⁻¹, respectively (Table 4). Generally, the high Ni dispersion and *S_m* could result in the high catalytic activity, which might be the reason for the best catalytic performance in Fig. 2 and Fig. 3.

In Fig. 9d, the CO₂-TPD profiles of the three catalysts displayed two CO₂ desorption peaks. The peak in the range of 110 – 183 °C attributed to the weak basic site on the catalyst surface with the monodentate coordination of CO₂ [37]; while the other peak belonged to the moderate basic site with the bidentate coordination of CO₂ [37,38]. Interestingly, the peaks of the weak and moderate basic sites both shifted to low temperature as the increase of Mg species content, indicating more Mg species could reduce the activation difficulty of CO₂. Inexplicably, the peak areas of the CO₂-TPD profiles of Ni₈Mg₂-NT and Ni₅Mg₅-NT were similar, which were much larger than that of Ni-NT. In all, the doping of Mg species led to enhancement of both CO₂ adsorption and activation capacity.

Table 4

H₂ uptake, Ni dispersion, active surface area and TOF of the 600 °C-reduced catalysts.

Samples	H ₂ uptake (μmol·g ⁻¹) ^a	<i>D</i> ^b (%)	<i>S_m</i> ^c (m ² ·g ⁻¹)	TOF _{CO₂} 220 °C (s ⁻¹) ^d
Ni-NT	255.6	8.1	5.2	1.09 × 10 ⁻³
Ni ₈ Mg ₂ -NT	425.4	16.5	19.1	2.98 × 10 ⁻³
Ni ₅ Mg ₅ -NT	357.6	10.3	12.5	2.89 × 10 ⁻³

^a H₂ uptake, calculated based on the H₂-TPR and H₂-TPD results.

^b Ni dispersion, calculated from the H₂-TPR and H₂-TPD results.

^c Active surface area

^d Turnover frequency, calculated based on the metal dispersion and the CO₂ conversion at 220 °C.

3.6. XPS analysis

Fig. 10a showed the wide XPS spectra of the three 600 °C-reduced catalysts. Besides the peaks of Ni, Si and O, the peak of the Mg 1s was also seen over Ni₈Mg₂-NT and Ni₅Mg₅-NT. Moreover, the peak intensities of Mg species (around 1305.3 and 307.3 eV) were directly proportional to the Mg content. For the Ni 2p_{3/2} XPS spectra, the bonding energy of Ni⁰ of the three reduced Ni-NT, Ni₈Mg₂-NT, and Ni₅Mg₅-NT catalysts was the same of 852.6 eV, suggesting that the Mg modification did not alter the electronic environment of metallic Ni

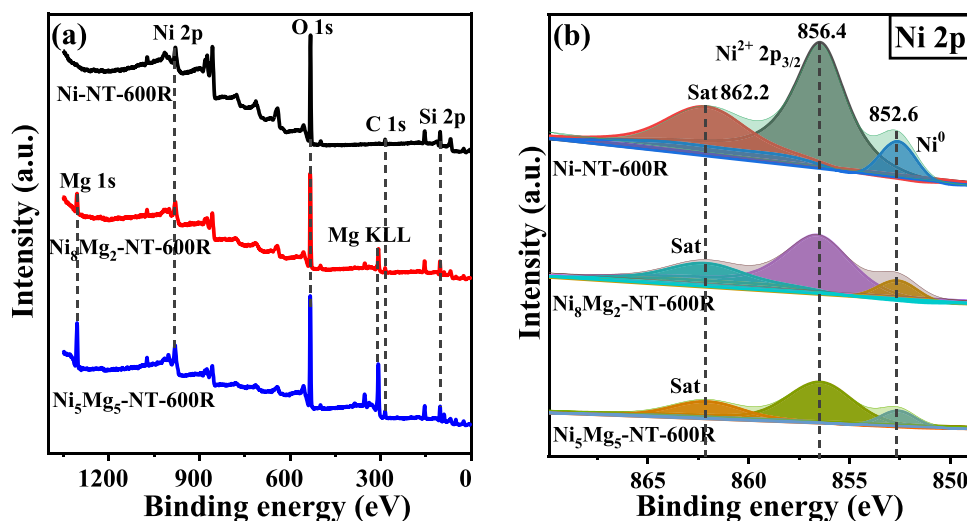


Fig. 10. Wide XPS spectra (a) and Ni 2p spectra (b) of the 600°C-reduced catalysts.

(Fig. 10b). Furthermore, the peaks of 856.4 eV and 862.2 eV on the Ni 2p_{3/2} spectra were attributed to Ni²⁺. As mentioned above, the reduction degree of Ni₈Mg₂-NT-600R was 96.6%; thus, the Ni²⁺ species in Fig. 10b was derived from the oxidation of metallic Ni in air as well as the unreduced Ni-phyllsilicate species [39].

3.7. Long-term stability test

To study the long-term stability of the optimal Ni₈Mg₂-NT catalyst, an operating temperature of 450 °C was chosen for the 100-h test of CO₂ methanation. Meanwhile, Ni-NT was also evaluated as the reference. Both the 600°C-reduced Ni-NT and Ni₈Mg₂-NT catalysts did not reach thermodynamic equilibrium at 450 °C, and the change of the catalyst surface structure was sensitive with its catalytic activity. As shown in Fig. 11, the CO₂ conversion decreased by approximately 1.4% for Ni₈Mg₂-NT and 1.8% for Ni-NT. The CH₄ selectivity decreased by approximately 0.2% for Ni₈Mg₂-NT and 0.4% for Ni-NT. After the test, the spent catalysts were characterized by XRD and TEM (Fig. S6). The nickel crystal size of the spent Ni-NT increased to 5.41 nm (Fig. S6a) with the average Ni particle size of 5.87 nm (Fig. S6b); at the same time, the nickel particle size of the spent Ni₈Mg₂-NT was 5.24 nm (Fig. S6c) with the Ni crystal size of 4.92 nm (Fig. S6d). Apparently, Ni₈Mg₂-NT exhibited the higher Ni sintering resistance. In short, the both Ni-NT and Ni₈Mg₂-NT catalyst displayed high long-term stability, and Ni₈Mg₂-NT

seemed to be much more stable than Ni-NT.

4. Discussion

4.1. Effect of Ni/Mg ratio and reduction temperature

As mentioned in Fig. 2 and Fig. S1, both Ni/Mg ratio and reduction temperature played a significant role in boosting the catalytic activity. As shown in Fig. 12, the nanotubular materials could be successfully formed with the Ni/Mg ratio varying from 10:0–0:10, whose color changed with the component; while the crystal phase was still 1:1 type phyllsilicate with the formula of M₃Si₂O₅(OH)₄ (M=Ni or Mg). However, the Ni content as well as Mg doping were both important for the catalytic activity, which followed the sequence of Ni₈Mg₂-NT > Ni₅Mg₅-NT > Ni-NT > Ni₂Mg₈-NT at the reduction temperature of 700 °C for CO₂ methanation. Thus, the Ni/Mg ratio seemed to be more important than the Ni content. The formation of Ni-Mg phyllsilicate solid solution and the intimate contact with Ni species and Mg species could maximize the promotion effect, and Ni₈Mg₂-NT obtained higher catalytic activity compared with Ni-NT. Furthermore, the Ni/Mg ratio was closely related with the anti-collapse property. During the reduction process, Ni-phyllsilicate would convert to Ni/SiO₂ catalyst in H₂ flow resulting in the destruction and collapse of the tubular structure; however, the thermal stability and nonreducible property of Mg-phyllsilicate species

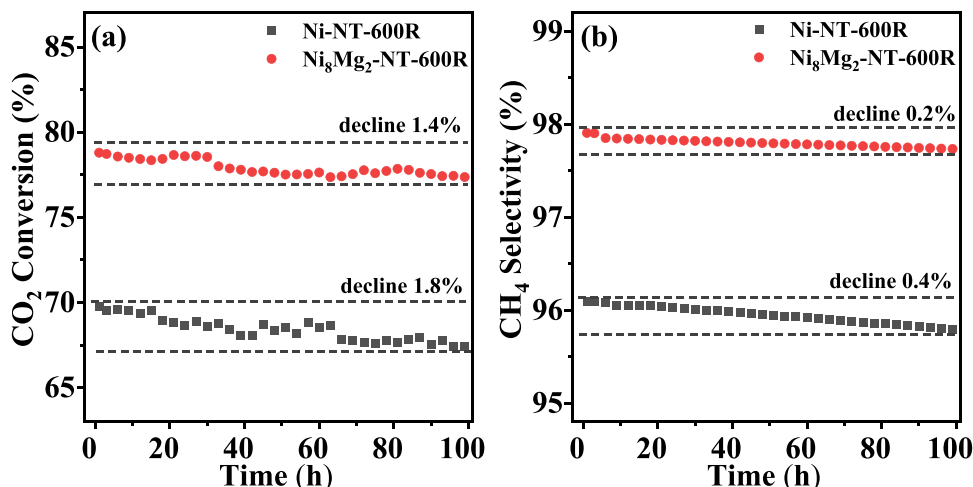


Fig. 11. Long-term stability test of the 600°C-reduced Ni-NT and Ni₈Mg₂-NT: (a) CO₂ conversion, and (b) CH₄ selectivity.

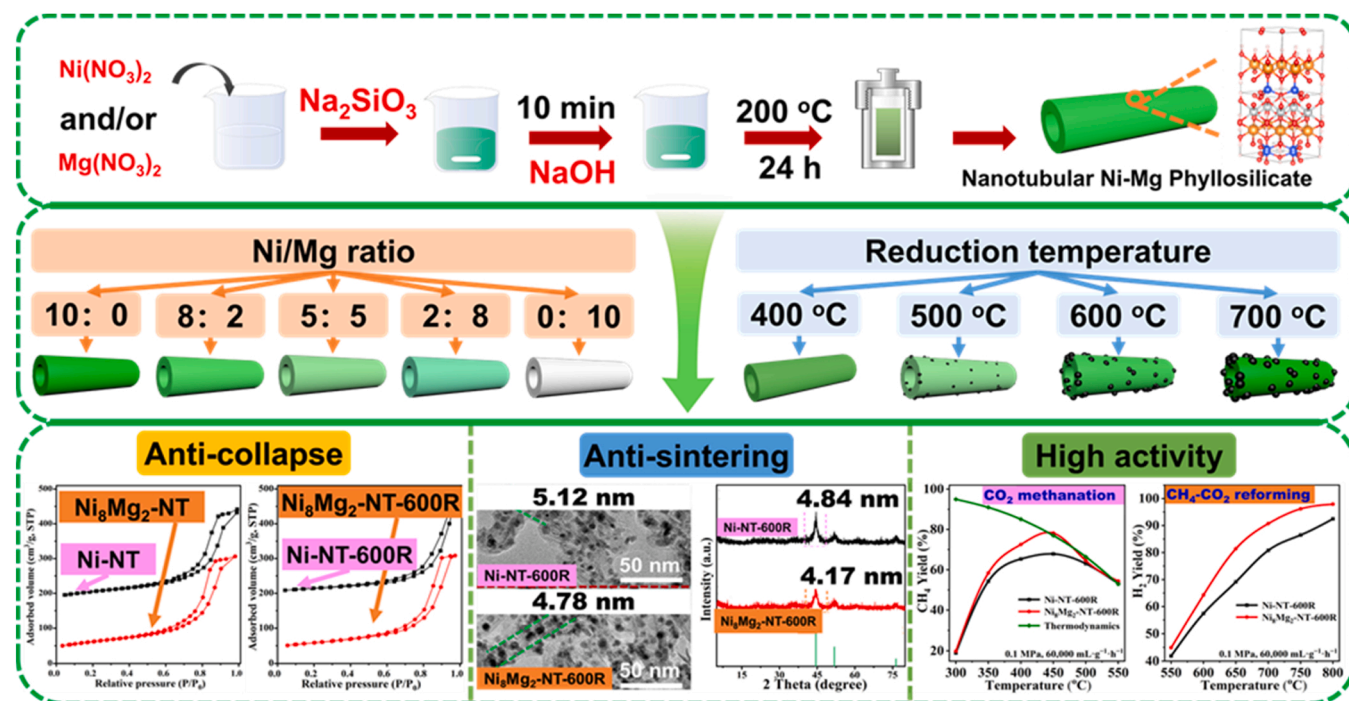


Fig. 12. Overview of the effect of Ni/Mg ratio and reduction temperature on physical and chemical properties of the nanotubular Ni-Mg bimetallic phyllosilicate.

could retain its original form to provide the support and physical barrier for Ni/SiO₂, leading to the enhancement of the anti-collapse of the nanotubes.

On the other hand, the stability and anti-collapse property could be maintained below a certain operation temperature, and the too high temperature was also the challenge for the Ni-Mg phyllosilicate materials. Thus, the reduction temperature was also optimized in this work, and 600 °C was the optimal value. This was because the too high temperature (700 °C) led to the serious collapse and the passivation of the active centers, and too low temperature (400 and 500 °C) could not achieve the enough active Ni species. As a result, the optimal 600 °C-reduced-Ni₈Mg₂-NT catalyst showed the highest CO₂ conversion compared to some reported CO₂ methanation catalysts in literature (Table S3). Meanwhile, this catalyst still exhibited the high CH₄ conversion for CH₄-CO₂ reforming (Table S4). In all, the modification of Mg species to for the nanotubular Ni-Mg phyllosilicate solid solution could improve the anti-collapse, anti-sintering and catalytic activity compared with the nanotubular Ni-phyllosilicate.

4.2. Kinetic analysis for CO₂ methanation

In order to further explore the effect of doped species Mg on catalytic performance, the kinetic analysis of the 600 °C-reduced Ni-NT, Ni₈Mg₂-NT and Ni₅Mg₅-NT for CO₂ methanation was performed. In this test, CO₂ conversion was controlled to below 10% using different space velocities and operation temperatures (210, 220, and 230 °C), and the activation energy (E_a) was calculated following the Arrhenius equation (Fig. 13). The order of E_a values for the three catalysts was Ni-NT (143.25 kJ·mol⁻¹) > Ni₅Mg₅-NT (103.26 kJ·mol⁻¹) > Ni₈Mg₂-NT (55.29 kJ·mol⁻¹). This result was in good agreement of the catalytic activity trend in Fig. 2. Furthermore, the calculated TOF_{CO2} (220 °C) values for Ni-NT, Ni₈Mg₂-NT and Ni₅Mg₅-NT were 1.09×10^{-3} , 2.98×10^{-3} and 2.89×10^{-3} s⁻¹, respectively (Table 4). The similar TOF_{CO2} of Ni₈Mg₂-NT and Ni₅Mg₅-NT indicated that the intrinsic catalytic power of each active site was similar, and the different Ni content led to the different catalytic activity between them. On the other hand, the low TOF_{CO2} of Ni-NT was the root cause for its low catalytic activity. In short, the addition of the species Mg had an important effect on the

both E_a and TOF_{CO2} of the catalyst due to differences in nickel content, nickel particle size, and the chemisorption properties of H₂ and CO₂.

4.3. Reaction pathway and intermediate analysis by in-situ DRIFTS

To study the possible pathways and reaction intermediates of the CO₂ methanation reaction catalyzed by the Ni-NT and Ni₈Mg₂-NT catalysts, the *in-situ* DRIFTS analysis was performed (Fig. 14). For the spectra of Ni-NT in Fig. 13a, the absorption peaks around 3750 cm⁻¹ belonged to the O–H stretching vibration of the Si–OH. Furthermore, the absorption peak intensity of the O–H stretching vibrations gradually increased as the temperature varied from 50 to 350 °C, indicating a gradual increase in the amount of O–H with increasing temperature. This was derived from the formation of H₂O or OH in the CO₂ methanation reaction on the catalyst surface [40–42]. In addition, the peaks around 3010 and 1302 cm⁻¹ attributing to C–H stretching vibration were also observed, which usually consisted of methylene (=CH–), methyl (CH₃–), and –CH₂– groups [43]. All of them were the key reaction intermediates in the hydrogenation of CO₂ to CH₄, and could provide the important evidence for the formation of CH₄. Significantly, in the enlarged view with the wavenumber range of 2960–3060 cm⁻¹, the C–H peak occurred from 225 °C, which could be considered as its temperature for initiation of activity. Additionally, the observed peaks at 2390 and 2320 cm⁻¹ corresponded to the adsorbed CO₂ molecule on the catalyst surface. Interestingly, the absorbance peaks at 1930 and 1830 cm⁻¹ were observed above 150 °C, which should be caused by the stretching vibration of the bridged CO molecule – one of the reaction intermediates for methanation of CO₂ [44]. Furthermore, the characteristic absorbance peak of monodentate HCOO⁻ was located at 1580 cm⁻¹, indicating the CO₂ methanation reaction also underwent the reaction intermediate of HCOO⁻ [45]. Similar with the peaks of CO, the HCOO⁻ peak also occurred at 150 °C, indicating the formation of intermediates of CO and HCOO⁻ was synchronous.

On the other hand, the similar absorbance peaks could also be seen on the *in-situ* DRIFTS spectra of the 600 °C-reduced Ni₈Mg₂-NT catalyst (Fig. 14b). It was worth noting that the two intermediates of CO and HCOO⁻ occurred at a lower temperature of 125 °C, and their peak intensities were also stronger than those of Ni-NT. Meanwhile, the

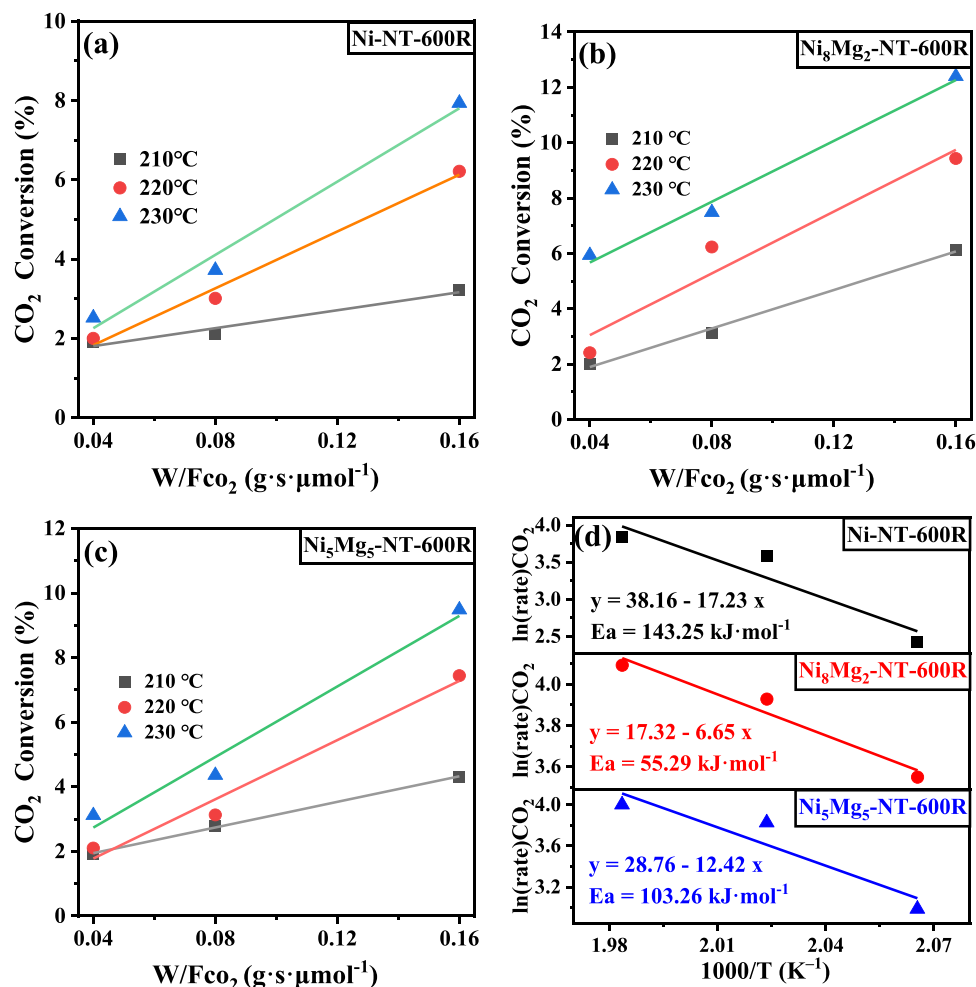
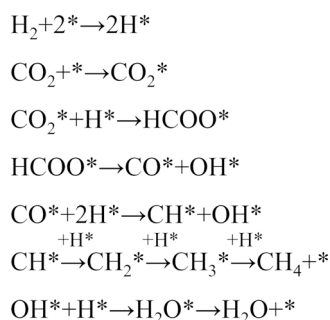


Fig. 13. Variation of CO₂ conversion of the 600 °C-reduced catalysts with W/Fco₂: (a) Ni-NT, (b) Ni₈Mg₂-NT, and (c) Ni₅Mg₅-NT; and Arrhenius plots for CO₂ methanation on different catalysts (d).

absorbance peak of C–H bond appeared at 200 °C with relatively strong intensity, which was also lower than that of Ni-NT. The above results indicated that the Ni₈Mg₂-NT catalyst possessed better low-temperature catalytic activity than Ni-NT. Finally, it was concluded that mono-dentate HCOO⁻ and bridged CO were possible reaction intermediates for further hydrogenation to CH_x – species for the CO₂ methanation [41]. The reaction pathway inferred from *in-situ* DRIFTS was shown below:



5. Conclusions

To suppress the problems of easy collapse, Ni sintering and low

catalytic activity of the nanotubular nickel phyllosilicate, a series of Mg-doped nanotubular nickel-magnesium phyllosilicate catalysts were prepared by a simple hydrothermal method, and evaluated for both CO₂ methanation and CH₄-CO₂ reforming reactions. In order to obtain the optimal catalyst, the Ni/Mg ratio as well as the reduction temperature were investigated, and the 600 °C-reduced Ni₈Mg₂-NT catalyst with Ni/Mg ratio of 8:2 showed high catalytic performance in CO₂ methanation with a CO₂ conversion of 78.8% and CH₄ selectivity of 99.1% at 450 °C, 0.1 MPa, 60000 mL·g⁻¹·h⁻¹, low E_a of 55.29 kJ·mol⁻¹ and large TOF_{CO₂} of $2.98 \times 10^{-3} \text{ s}^{-1}$ at 220 °C. *In situ* DRIFTS results displayed that CO and HCOO⁻ were the key intermediates for hydrogenation of CO₂ to CH₄, and the occurrence of these intermediates and CH_x – species at lower temperature further testified the higher activity of Ni₈Mg₂-NT. Additionally, this catalyst also should high catalytic activity for CH₄-CO₂ reforming with the CO₂ conversion of 95.8% and CH₄ conversion of 97.9% at 800 °C, and high long-term stability for CO₂ methanation with excellent anti-sintering property. The characterization results showed that the formed nanotubular Ni-Mg bimetallic phyllosilicate was in the state of solid solution rather than physical mixture of Ni-phyllosilicate and Mg-phyllosilicate. The high thermal stability and nonreducible property of Mg-phyllosilicate species as well as its intimate contact with Ni-phyllosilicate resulted in the enhanced the anti-collapse and anti-sintering properties of the Ni-Mg phyllosilicate compared with the conventional tubular Ni-phyllosilicate. The decreased Ni particle size and increased H₂ and CO₂ chemisorption capacity resulted in the boosting catalytic performance. In all, the construction of the nano-tubular Ni-Mg bimetallic phyllosilicate could address the problems of

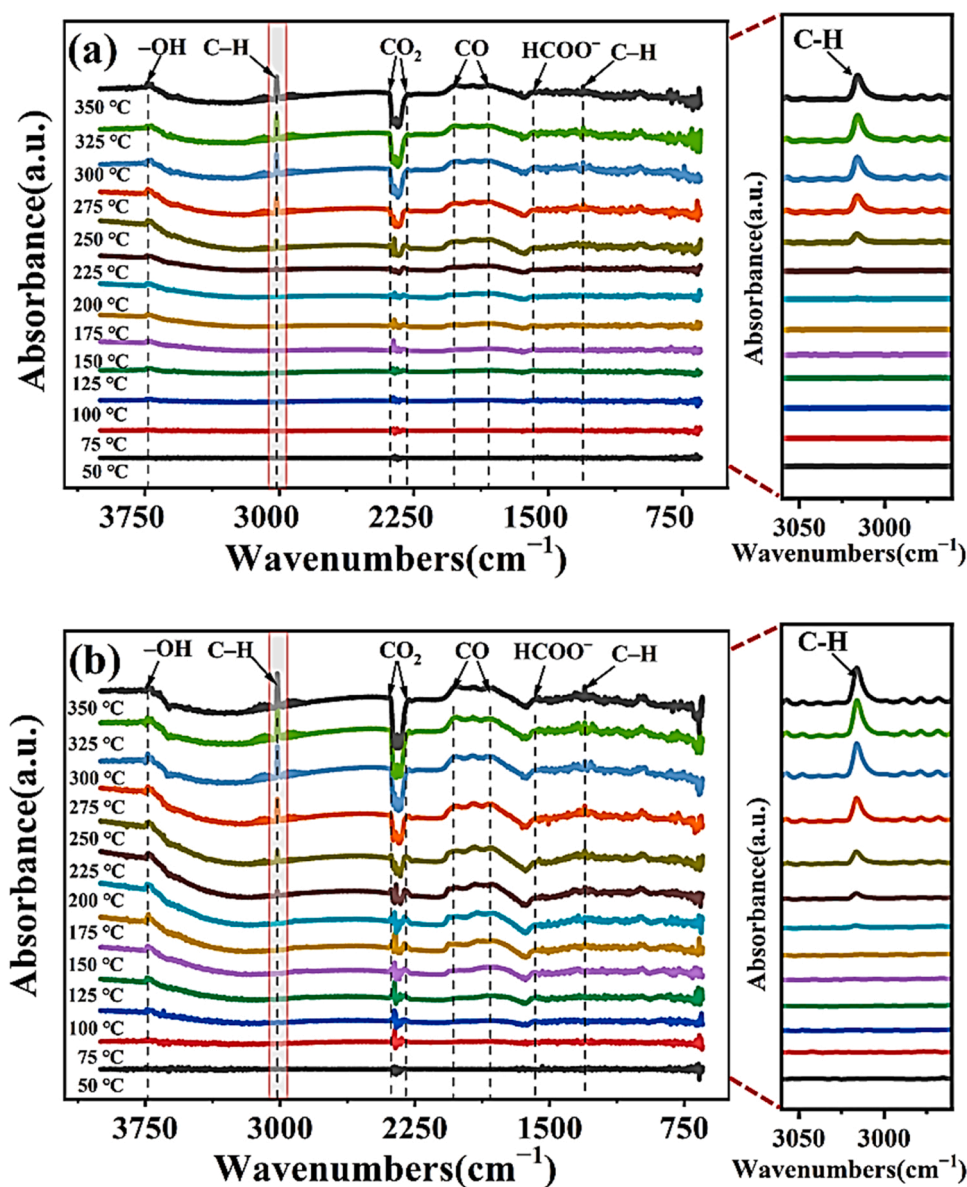


Fig. 14. *In-situ* DRIFTS analysis for CO₂ methanation of the 600 °C-reduced catalysts: (a) Ni-NT and (b) Ni₈Mg₂-NT.

the conventional nanotubular Ni-phyllsilicate due to its solid solution structure and the special characteristics of Ni-Mg-phyllsilicate species.

CRediT authorship contribution statement

Dehui Wang: Writing – original draft, Characterization, Data curation, Validation. **Jia Liu:** Writing – original draft, Investigation, Data curation. **Hai Li:** Data curation. **Qing Liu:** Writing – review & editing, Supervision, Methodology, Project administration. **Yu Cheng:** Writing – review & editing. **Xing Fan:** Writing – review & editing. **Peng Liang:** Writing – review & editing.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data Availability

Data will be made available on request.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.apcatb.2023.122452](https://doi.org/10.1016/j.apcatb.2023.122452).

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